



# Large magnetocaloric effect in TbCo<sub>3</sub>B<sub>2</sub> compound

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## ABSTRACT

A single-phased TbCo<sub>3</sub>B<sub>2</sub> compound has been prepared by an arc-melting method. The magnetic properties and magnetocaloric effect (MCE) have been studied by the magnetization and heat capacity measurements. A large reversible magnetocaloric effect has been observed accompanied by a second-order phase transition at around Tb–Tb sublattice ordering temperature. The values of maximum magnetic entropy change ( $-\Delta S_M^{\max}$ ) reach 4.9 and 10.3 J kg<sup>-1</sup> K<sup>-1</sup> for the field change of 2 and 7 T with no obvious hysteresis loss around 30 K, respectively. The corresponding maximum adiabatic temperature changes ( $\Delta T_{\text{ad}}^{\max}$ ) are evaluated to be 4.0 and 8.6 K. The magnetic transition and the origin of large MCE in TbCo<sub>3</sub>B<sub>2</sub> were discussed.

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## 1. Introduction

Recently, the magnetocaloric effect (MCE) in various magnetic materials has been extensively studied experimentally and theoretically, not only because of their great potential for magnetic refrigeration applications but also for further understanding the fundamental physical properties of the materials [1–11]. The MCE is a magnetothermodynamic phenomenon, it manifests as an isothermal magnetic entropy change ( $\Delta S_M$ ) or an adiabatic temperature change ( $\Delta T_{\text{ad}}$ ) when the magnetic material is exposed to a varying magnetic field. Magnetic refrigeration based on the MCE is advantageous being an environment friendly and energy efficient refrigeration mechanism, which is expected to be an important future cooling technology [1–4]. A large value of MCE is considered to be the most important requirement of the application, and therefore it is desirable to find new materials with large MCE especially at low magnetic fields and with a wide temperature range. Some rare-earth based compounds with a ferromagnetic (FM) or an antiferromagnetic (AFM) phase transition have been found to possess not only large magnetic entropy change but also a small hysteresis loss.

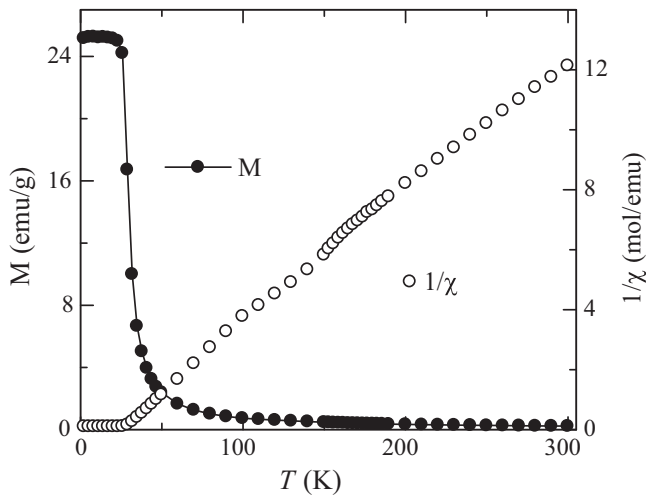
The ternary intermetallic compounds of the RCo<sub>3</sub>B<sub>2</sub> where R is a rare earth or yttrium have attracted some attentions due to their interesting physical properties [12–15]. The RCo<sub>3</sub>B<sub>2</sub> compounds are paramagnetic at room temperature and undergo a Co–Co sublattice magnetic order transition at ~160 (20) K. In addition, a second R–R sublattice magnetic order transition was also observed at 54, 47, 28 and 22 K for R=Gd, Sm, Tb and Dy, respectively [12–15]. Very recently, a giant/large reversible MCE in ternary R–Co–B intermetallic compounds RCo<sub>2</sub>B<sub>2</sub> (R=Gd, Pr and Nd) [16–18] was reported, respectively. To search new material displaying large MCE and further understand the physical properties of RCo<sub>3</sub>B<sub>2</sub> system, in this paper, the magnetic properties and MCE in TbCo<sub>3</sub>B<sub>2</sub> were systematically studied.

## 2. Experimental

The polycrystalline TbCo<sub>3</sub>B<sub>2</sub> was synthesised by an arc melting method using a tungsten electrode under an argon atmosphere. Stoichiometric amounts of high purity Tb, Co and B were melted more than six times for homogeneity on a water-cooled copper hearth. The total weight loss of the sample in this step was less than 0.5%. Then the samples were finally annealed at 1073 K for one week in evacuated quartz tubes. The sample was proved to be single phase with the CeCo<sub>3</sub>B<sub>2</sub>-type hexagonal crystal structure belonging to the *P6/mmm* space group by X-ray diffraction (XRD) using Rigaku RINT 2250 diffractometer. The lattice parameters *a* and *c* were evaluated to be 5.047 and 3.003 Å, respectively, from XRD data using Rietveld refinement method. The magnetization measurements were done using a superconducting quantum interference device magnetometer (Quantum Design, MPMS-7) in the temperature range 2–300 K, with DC magnetic fields from 0 to 7 T. The specific heat measurements were carried out by the adiabatic heat relaxation method in the temperature range from 2 to

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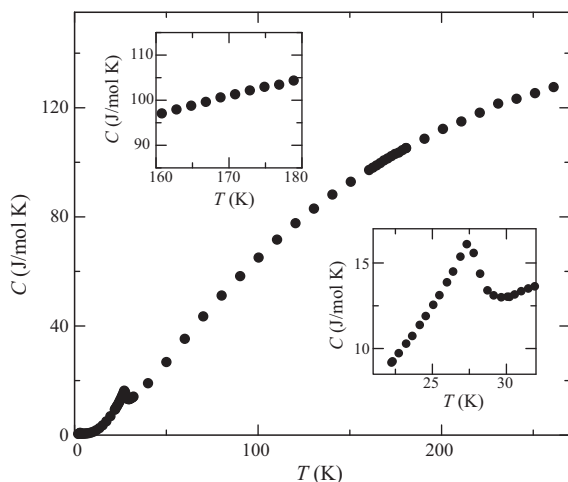
**Fig. 1.** Temperature dependence of magnetization ( $M$ ) and inverse susceptibility ( $1/\chi$ ) for  $\text{TbCo}_3\text{B}_2$  in an external magnetic field  $H = 0.1$  T.

300 K using a physical property measurement system (PPMS-9) from Quantum Design.

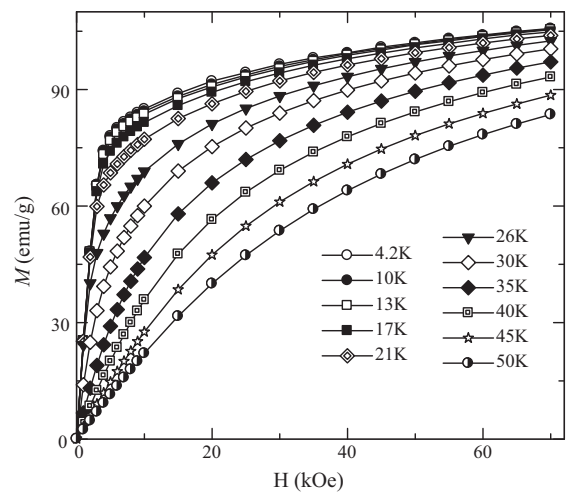
### 3. Results and discussion

The temperature dependence of magnetization ( $M$ ) and the reciprocal susceptibility ( $1/\chi$ ) for  $\text{TbCo}_3\text{B}_2$  in an external magnetic field  $H = 0.1$  T are shown in Fig. 1. A sharp transition around 30 K and the anomaly change around 170 K were observed. The reciprocal susceptibility shows Curie–Weiss behaviour above 200 K. Dubman et al. [14] studied the magnetic ordering of  $\text{TbCo}_3\text{B}_2$  by means of neutron diffraction method, and they concluded that the magnetic ordering transition of the Tb sublattice ( $\sim 30$  K) accompanied by the rotation of the magnetic moment towards the basal plane, the anomaly ( $\sim 170$  K) was due to the Co–Co sublattice magnetic ordering transition. The temperature dependence of zero field specific heat results for  $\text{TbCo}_3\text{B}_2$  is shown in Fig. 2, a clear  $\lambda$ -shape indicating a magnetic transition around 28 K which is a typical character of second order phase transition, and no obvious change can be observed around 170 K. These behaviours were consistent with the magnetic measurements and previous reported results [14,15].

A set of magnetic isotherms on increasing and decreasing field were measured for  $\text{TbCo}_3\text{B}_2$  with the temperature range from 4 to



**Fig. 2.** Temperature dependence of zero field specific heat ( $C$ ) for  $\text{TbCo}_3\text{B}_2$ . Inset show the  $C(T)$  curves for  $\text{TbCo}_3\text{B}_2$  around 28 K and 170 K, respectively.

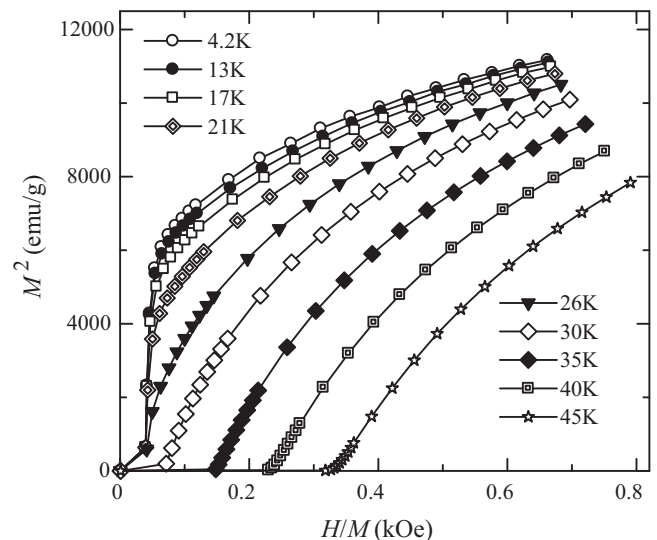


**Fig. 3.** Magnetic field dependence of the magnetization for  $\text{TbCo}_3\text{B}_2$  at some selected temperatures up to 7 T.

60 K up to 7 T. There is no obvious hysteresis for  $T > 10$  K. To ensure the readability of the figure, only several isotherms with increasing field are presented in Fig. 3. For low temperature ones, the magnetization  $M$  tends to be saturated at low field. A large reversible MCE is expected around the transition temperature where the magnetization rapidly changes with varying temperature. Based on the Inoue-Shimizu model, which involves a Landau expansion of the magnetic free energy up to sixth power of the total magnetization  $M$ , can be used to determine the transition type [19],

$$F(M, T) = \frac{c_1(T)}{2}M^2 + \frac{c_2(T)}{4}M^4 + \frac{c_3(T)}{6}M^6 + \dots - BM \quad (1)$$

The parameters  $c_1(T)$ ,  $c_2(T)$  and  $c_3(T)$  represent the Landau coefficient, and it has been reported that the order of a magnetic transition is related to the sign of the  $c_2(T)$ . A transition is expected to be the first order when  $c_2(T_C)$  is negative, whereas it will be the second order for a positive  $c_2(T_C)$ . The sign of  $c_2(T_C)$  can be determined by means of Arrott plots [20,21]. If the Arrott plot is S-shaped near  $T_C$ ,  $c_2(T_C)$  is negative, otherwise, positive. To further understand the nature of the magnetic transition in  $\text{TbCo}_3\text{B}_2$ , the Arrott-plots  $M^2$  vs.  $H/M$  at some selected temperatures for  $\text{TbCo}_3\text{B}_2$  are plotted in Fig. 4. Neither the inflection point nor negative slopes



**Fig. 4.** The Arrott-plot of  $\text{TbCo}_3\text{B}_2$  at some selected temperatures.

is observed. Together with the  $\lambda$ -shape in  $C(T)$  (as shown in Fig. 2) the present results confirmed that the  $\text{TbCo}_3\text{B}_2$  compound undergoes a second order magnetic phase transition from paramagnetic to ferromagnetic state.

According to thermodynamical theory [22], the isothermal magnetic entropy changes associated with a magnetic field variation is given by

$$\Delta S_M(T, \Delta H) = S_M(T, H) - S_M(T, 0) = \mu_0 \int_0^{H_{\max}} \left( \frac{\partial S(H, T)}{\partial H} \right)_T dH \quad (2)$$

From the Maxwell's thermodynamic relation:

$$\left( \frac{\partial S(H, T)}{\partial H} \right)_T = \left( \frac{\partial M(H, T)}{\partial T} \right)_H \quad (3)$$

one can obtain the following expression:

$$\Delta S_M(T, \Delta H) = \mu_0 \int_0^{H_{\max}} \left( \frac{\partial M(H, T)}{\partial T} \right)_H dH \quad (4)$$

where  $S, M, H$ , and  $T$  are the magnetic entropy, magnetization of the material, applied magnetic field, and the temperature of the system, respectively. From the magnetization measurements made at discrete field and temperature intervals,  $\Delta S_M$  can be approximately calculated by the following expression:

$$\Delta S_M(T, \Delta H) \approx \frac{1}{\delta T} \left[ \mu_0 \int_0^{H_{\max}} M(T + \delta T, H) dH - \mu_0 \int_0^{H_{\max}} M(T, H) dH \right] \quad (5)$$

Based on the method described by Pecharsky and Gschneidner [22], the accuracy of the  $\Delta S_M$  calculated from the magnetization data for the materials studied here is better than 10%. The  $\Delta S_M$  was calculated using Eq. (5) in the vicinity of its ordering temperature based on the results of magnetization isotherms. The resulting changes of magnetic entropy  $-\Delta S_M$  as a function of temperature for different magnetic field variations up to 7 T are shown in Fig. 5. A large magnetocaloric effect can be observed around 30 K. The maximum values of magnetic entropy change ( $-\Delta S_M^{\max}$ ) reach 8.7 and  $10.3 \text{ J kg}^{-1} \text{ K}^{-1}$  for a field change of 5 and 7 T, respectively. This large MCE is related to a second order Tb–Tb sublattice magnetic phase transition. Another important parameter for a MCE material is the temperature dependence of adiabatic temperature change  $\Delta T_{\text{ad}}$ , which was also evaluated. Keeping this context in mind, the temperature dependence of  $\Delta T_{\text{ad}}$  (shown in Fig. 6) for various magnetic field changes up to 7 T has been calculated using the  $\Delta S_M(T)$  and zero-field specific heat results (Fig. 2). The overall nature of  $\Delta T_{\text{ad}}$  as a function of temperature is remarkably similar to that of  $\Delta S_M(T)$ . The maximum values of adiabatic temperature

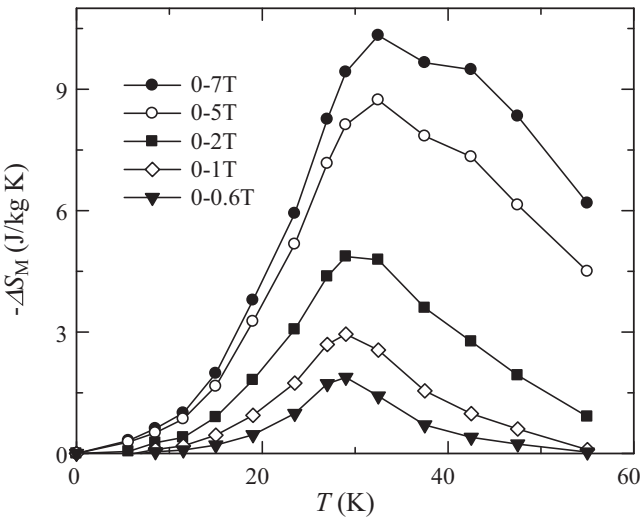


Fig. 5. The magnetic entropy change  $-\Delta S_M$  as a function of temperature for various magnetic field changes up to 7 T for  $\text{TbCo}_3\text{B}_2$ .

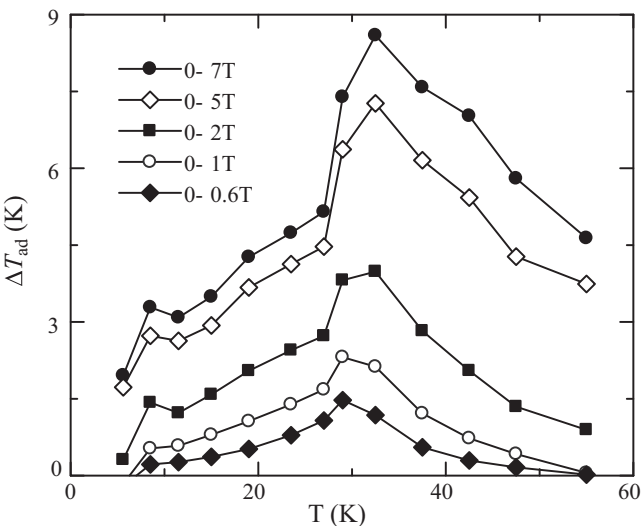


Fig. 6. The adiabatic temperature change  $\Delta T_{\text{ad}}$  as a function of temperature for various magnetic field changes up to 7 T for  $\text{TbCo}_3\text{B}_2$ .

change ( $\Delta T_{\text{ad}}^{\max}$ ) reach 7.3 and 8.6 K for a field change of 5 and 7 T, respectively. The transition temperature ( $T_M$ ), the order of magnetic transition type, the maximum values of magnetic entropy change  $-\Delta S_M^{\max}$  and adiabatic temperature change  $\Delta T_{\text{ad}}^{\max}$  as well as the relative cooling power [RCP, which is defined as the product of as the product of the maximum magnetic entropy change

**Table 1**  
The transition temperature ( $T_M$ ), the order of transition type as well as the maximum values of magnetic entropy change  $-\Delta S_M^{\max}$  and adiabatic temperature change  $\Delta T_{\text{ad}}^{\max}$  as well as the relative cooling power (RCP) under 2 and 5 T for  $\text{TbCo}_3\text{B}_2$  and some MCE materials.

Material	$T_M$ (K)	Order of transition	$-\Delta S_M^{\max} (\text{J kg}^{-1} \text{ K}^{-1})$		RCP ( $\text{J/cm}^3$ )		$\Delta T_{\text{ad}}^{\max}$ (K)		Ref.
			2 T	5 T	2 T	5 T	2 T	5 T	
$\text{TbCo}_3\text{B}_2$	28	Second	4.9	8.7	0.5	2.5	4.0	7.3	Present
$\text{PrCo}_2\text{B}_2$	16	Second	4.9	8.1	0.36	1.1	4.3	8.1	[17]
$\text{NdCo}_2\text{B}_2$	27	Second	4.5	7.1	0.3	0.8	3.3	5.8	[18]
$\text{DySb}$	11	First	–	15.8	–	~1.4	–	–	[23]
$\text{GdCo}_2\text{B}_2$	25	First	9.3	17.1	1.06	3.8	6.7	15.4	[16]
$\text{Ho}_5\text{Pd}_2$	28	First	~7	~18	1.7	6.3	–	–	[19]
$\text{GdPd}_2\text{Si}$	17	Second	4.5	15	~0.4	~2.3	3.2	8.5	[24]
$\text{PrNi}$	20	Second	2.4	6.1	~0.2	~0.7	0.8	1.7	[25]
$\text{ErCo}_2$	35	First	28	33	1.5	3.08	4.2	9.5	[2,5]

$\Delta S_M^{\max}$  and full width at half maximum  $\delta T_{\text{FWHM}}$ ] under 2 and 5 T for  $\text{TbCo}_3\text{B}_2$  and some MCE materials with  $T_M$  around 25 K are listed in Table 1. Despite the fact that the  $\Delta S_M^{\max}$ ,  $\Delta T_{\text{ad}}^{\max}$  and RCP values of  $\text{TbCo}_3\text{B}_2$  are smaller than those of which processed the first order magnetic transitions, the present values are comparable with those of which undergoing the second order magnetic transitions. However, compared with some first order magnetic transition materials, the present  $\text{TbCo}_3\text{B}_2$  compound and some second order magnetic transition materials have high reversible MCE which makes them to be competitive materials for active magnetic-refrigeration application.

#### 4. Conclusions

In summary, we have systematically studied the magnetic properties and magnetocaloric effect (MCE) of  $\text{TbCo}_3\text{B}_2$  by determining the magnetization and heat capacity. A sharp transition around 30 K and an anomaly change around 170 K has been observed which was corresponding to the Tb–Tb and the Co–Co sublattice magnetic order transition, respectively. A large reversible MCE has been observed which is related to a second-order Tb–Tb sublattice magnetic phase transition. The present results may give some clue for searching new materials with large MCE.

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